

Microstructures and Electrical Properties of HPMC/PVP Polymer Blend films complex with Ferric Chloride (FeCl_3)

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ABSTRACT

Microstructural studies on FeCl_3 doped Hydroxypropyl methyl cellulose (HPMC)/Poly vinyl pyrrolidone (PVP) blend films were carried out using X-Ray diffraction studies. The XRD data revealed that the crystalline regions of the HPMC/PVP blend film decreases up to a certain percentage of FeCl_3 and then increases. Electrical conductivity studies on these doped films suggest complex formation due to doping which affects microstructure and also ac conductivity of polymer films. All these results were analyzed and explained on the basis of micro structural modification of HPMC/PVP blends as function of dopant concentration.

Key words: XRD, AC Conductivity, Polymer Composites

INTRODUCTION

Nowadays polymer blending is an easy and fastest way of getting a new material suitable for a specific purpose. Another way of tailoring the polymeric properties of an individual polymer or blends is by doping. The dopant can induce modifications in the molecular network and microstructures of the polymers. When a transition metal salts like FeCl_3 and CoCl_2 are doped, brings about a significant change in the physical properties including micro structural parameters, optical, electrical, thermal, and mechanical properties. These changes in physical properties, depends on the chemical nature of the dopant and the way in which they interact with the host polymer¹⁻³ Various researchers used different techniques to understand the micro structural properties and electrical properties of doped polymer films. Some of the studies are: effect of Iodine doping on PVA⁴, temperature dependence of CuCl_2 doped PVA⁵, FeCl_3 and CrCl_3 doping effect on poly hydroxyl amine acid⁶, effect of NH_4I doping

on PVA⁷. Proton conducting polymers are of great importance as they are used in solid state electro chemical power sources, sensors, electro chromic displays and fuel cells^{8,9} The advantage of polymer electrolytes are their good mechanical properties, easy fabrication in to thin films of desirable size, and the ability to form good electrode/ electrolyte contact^{10, 11} The main challenge in this field is to fabricate a low cost films with good mechanical and good electrical properties.

As we mentioned earlier number of literatures are available on PVA, PEO, polymer electrolytes complexed with various salts. But we have not noticed any article regarding HPMC or HPMC/PVP blend based polymer electrolyte. Hence we are reporting here the potential use of HPMC/PVP blend films as a host for in-organic salts in the preparation of polymer electrolyte. Hydroxypropyl methyl cellulose (HPMC), a bio- polymer derivative of cellulose available in powder form, water soluble and forms films. PVP a water soluble, synthetic polymer

having good film forming nature. Both these were Microstructure parameters like crystallinity can be influenced by adding in-organic salts.

MATERIALS AND METHODS

HPMC and PVP powdered samples were purchased from Loba chemicals Mumbai. Pure blend and doped films were prepared by using solvent cast method^{12, 16} HPMC (5%wt) and PVP (2%wt) dissolved in 100ml of distilled water separately. After complete dissolution the solutions were filtered using filter paper to remove undissolved particles. 50ml of HPMC and 50ml of PVP were mixed and stirred continuously for 20 min with moderate speed. During stirring pre dissolved $FeCl_3$ salt of 0.2g was added to the blend solution and stirred continuously using magnetic stirrer for complete dissolution. After complete dissolution the solution was allowed for a while and then it is poured on to the clean glass plate and allowed to dry for a week. After complete dry, the film were peeled out of the glass plate and stored in desiccators to avoid moisture, following the same procedure other films of different concentration and salt were prepared. The prepared films appears to be yellowish transparent and they retains this transparency up to a certain percentage of the salt concentration in the blend matrix, after which they tend to be more opaque. This is an indication that the salt may possibly fully miscible creating a new polymer matrix constituted by one phase¹⁷.

EXPERIMENTAL PROCEDURE

X-ray diffraction measurements

The changes in the microstructure of the complexed films was examined by using Rigaku miniflex II diffractometer with Ni filtered *CuK α* radiation of wavelength 1.5406 Å, with graphite monochromator. The radiation was assessed at a voltage of 30KV and 15mA. The dried samples were scanned in a 2θ range 6° to 40°.

AC conductivity measurements

AC conductivity measurements were made using LCR Hi Tester Hioki (Japan) 3532-50 programmable computer interfaced digital LCR impedance meter in the frequency range of 50Hz to 5MHz in room temperature.

Mechanical Properties

Test for mechanical properties of these polymer composite films were carried out using Lloyd's Universal testing machine LRX Plug Model 5 KN. The obtained mechanical property parameters for these polymers are given in table 2.

RESULTS AND DISCUSSIONS

XRD studies

The pure HPMC/PVP blends and films of different $FeCl_3$ concentrations were prepared. The prepared samples were investigated using WAXS method. From the diffraction pattern obtained it is found that a film consists of both crystalline and amorphous regions. In blends one sharp peak at 12° and one broad peak at 20° are observed. In salt doped films sharp peak disappears and only one broad peak at $2\theta = 20^\circ$ was observed. The intensity of the peak decreases and width increases. This is

Table. 1: Microstructural parameters of HPMC/PVP blend films Doped with $FeCl_3$

Sample	2 θ in deg	d in Å	<N>	D=<N>.d in Å	g (%)	α^d	β	δ	α
HPMC/PVP	12.74	6.93	25.20	174.63	0.1	3.90E-05	6.64E-06	0.30	2.8
	20.43	4.34	3.08	13.36	0.1	1.70E-05	4.46E-05	0.06	3.5
HPMC/PVP +0.2% $FeCl_3$	13.13	6.73	31.90	214.68	0.2	4.4E-05	9.89E-09	0.13	2.20
HPMC/PVP +0.4% $FeCl_3$	20.27	4.37	1.75	7.61	0.3	7.85E-07	1.91E-06	0.38	4.64
HPMC/PVP +0.6% $FeCl_3$	21.02	4.22	2.26	9.50	0.2	1.47E-08	8.15E-08	0.05	5.06
HPMC/PVP +0.8% $FeCl_3$	21.07	4.21	3.01	12.70	0.3	8.42E-06	7.78E-05	0.06	7.49

due to the disruption of crystalline regions of the blends by the FeCl_3 . Crystallinity was calculated from the ratio of the area under the crystalline peaks and the respective total area under the peaks, using PEAKFIT4.1 software. Crystallinity decreases from 52.29(%) (Pure) to 21.30(%) (0.6g) and then increases to 26.52(%) for 0.8g salt doped film. Degree of crystallinity affects the electrical conductivity of the samples. The amorphous nature produces greater ionic diffusivity which increases

the ionic conductivity. The greater ionic conductivity can be obtained in amorphous polymers that have a fully flexible backbone¹⁸. Micro structural parameters obtained are tabulated in tables 1.

It is observed that lattice strain in all the samples is around 0.1%-0.3%, the statistical percentage of deviation of parameters arises due to computation is around 0.06-0.38. With these parameters as input we have further refined these

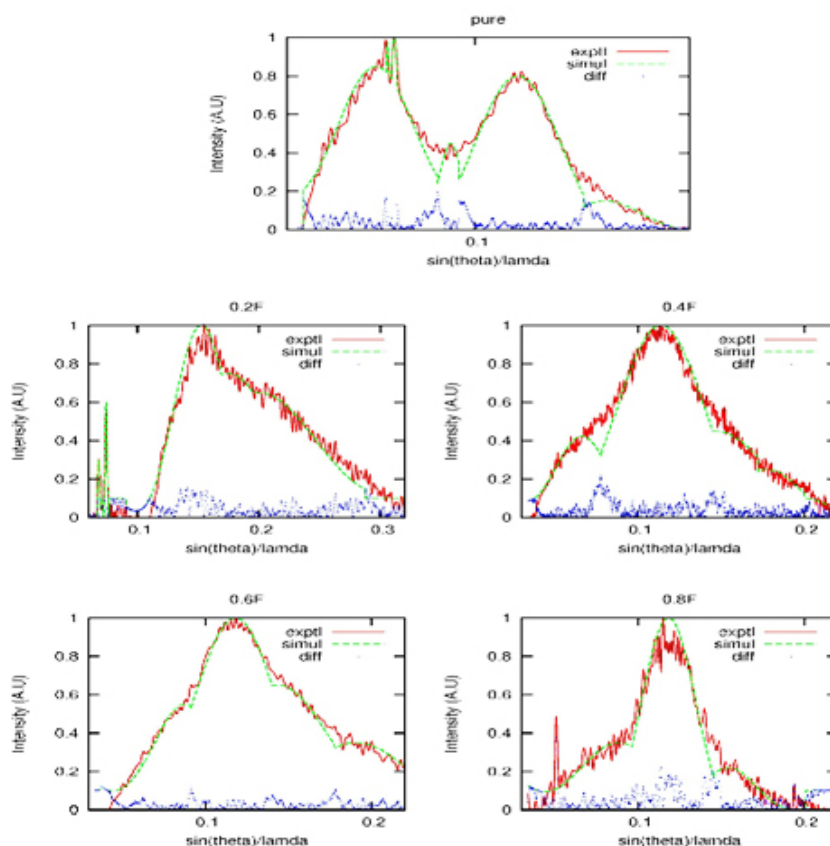


Fig. 1: WPPF of XRD patterns of the Pure HPMC/PVP and FeCl_3 doped blends

Table. 2: Mechanical properties of HPMC/PVP and HPMC/PVP: FeCl_3 polymer composites

Sample Elongation	Tensile Strength (MPa)	Stiffness (kN/m)	Young's Modulus (MPa)	% at Fracture
HPMC/PVP	42.21	-	1480.31	3.70
HPMC/PVP + 0.2% FeCl_3	12.74	945.85	1529.14	0.52
HPMC/PVP + 0.4% FeCl_3	10.71	23.90	286.88	23.54
HPMC/PVP + 0.6% FeCl_3	5.53	9.87	85.87	32.16
HPMC/PVP + 0.8% FeCl_3	40.27	79.60	1415.65	14.28

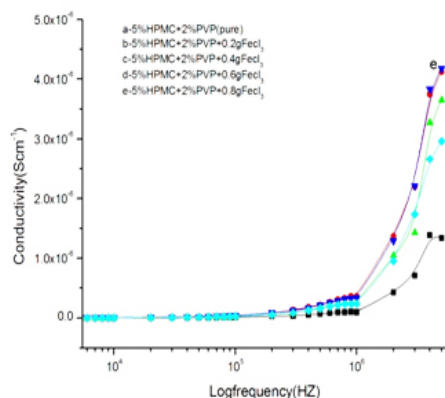


Fig. 2: Plot of conductivity versus log frequency

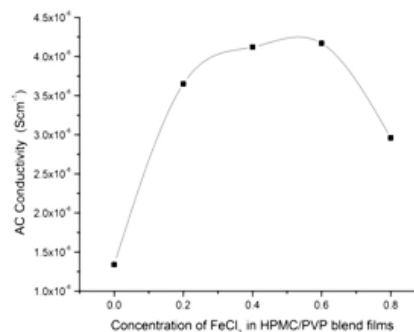


Fig. 3: Variation of conductivity at 1kHz as a function of concentration

parameters against whole pattern ($2\epsilon=6^\circ$ to 40°). The detailed theory of line profile analysis and whole powder pattern fitting method was discussed in our earlier papers¹⁹⁻²³. The goodness of the fit between simulated and experimental profiles for the samples given in figure 1. It is worth noting that none of the other parameters such as lattice strain, stacking fault probability varied much during refinement, against the whole pattern data of the samples. The deformation fault probability values given in the table 1, indicates that the values are low, because there are too many layers between two successive deformation fault layers. These observed facts confirm the complexation of salt with blend matrix. Background corrections and instrumental broadening corrections were carried out before doing these above said analysis.

AC conductivity

The average thickness and area of the prepared samples were found and, a piece of the sample was cut from the whole sample and silver paste was applied on both sides for good electrical contact between electrode-electrolyte. These samples were kept between the jaws of the sample holder for measurement of electrical conductivity and dielectric properties. Various parameters have been measured using Hioki (Japan) Model 3532-50 programmable computer interfaced digital LCR meter in the frequency range from 100 Hz to 5 MHz, at room temperature. The dielectric constant of a material can be calculated from the formula,

$$\epsilon' = cd / H \epsilon_0$$

where c is capacitance, d is thickness, ϵ_0 is the permittivity of free space, and A is the area of cross section of the sample and AC conductivity can be calculated from the equation below by which we have derived the conductivity for our samples.

$$\sigma = Gd / A \text{ S cm}^{-1}$$

Where G is the conductance, d is thickness, and A is the area of cross section of the sample. Frequency dependence of ac conductivity was given in Fig 2. It is observed that ac conductivity increases with frequency. The variation of ac conductivity with concentration of FeCl_3 is shown in the Fig 3. It is observed that the addition of FeCl_3 significantly improves the ionic conductivity. The conductivity of pure HPMC/PVP blend is ($1.34\text{E}-06$), highest conductivity is found to be ($4.17\text{E}-06$) for (0.6g) sample. Above 0.6g of FeCl_3 the conductivity decreases due to re-crystallization of the sample. The increase of conductivity is due to the increase of mobile charge carriers due to increase of salt concentration and also to the amorphous nature of the polymer electrolyte.

Mechanical Properties

We have studied the changes in the mechanical strength of these polymer composites

and the results obtained are shown in table 2. From the result it is seen that the tensile strength of the polymer composites decreases first with the concentration but it increased at the higher concentration. This indicates that the addition of FeCl_3 to the polymer, affects its mechanical property due to the breaking of bonds between the polymeric molecules. But later at the higher concentration, the FeCl_3 molecules form a network with the polymer and regain its strength.

CONCLUSIONS

X-ray diffraction study confirms that FeCl_3 disturbs the crystalline nature of HPMC/PVP blend

films. The increase in amorphous nature enhances the conductivity of the films. The highest conductivity found to be ($4.17\text{E-}06$ S/cm) for 0.6g FeCl_3 doped HPMC/PVP blend film, which has least crystallinity of 21.30 (%) (from XRD). AC conductivity studies confirm that the conductivity increases with salt concentration. These films with more flexibility, good mechanical strength and degradable nature are good polymer electrolyte for electro chemical devices. Even though the conductivity of these polymer composites are not so profound as compared to the normal conducting polymers²⁴, it is seen that there is an increase in the conductivity of upto three times than that of the pure PVA due to doping.

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