Analysis of AC-Conductivity in Chalcogenide $Ge_{10}Se_{20}$ Bi_{80}Thin Film

SHIVEOM SRIVASTAV

Department of Applied Science, Ambalika Institute Lucknow, India.

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ABSTRACT

The alloy $Ge_{10}Se_{20}Bi_{80}$ has been prepared. Thin films of $Ge_{10}Se_{20}Bi_{80}$ has been prepared via a thermal evaporation method (melt quenching technique) with 3000A thickness, and rate of deposition (4.1) A/sec at pressure $2x10^{-5}$ Torr. The A.C electrical conductivity of a- thin films $Ge_{10}Se_{20}$ Bi_{80} has been studied as a function of frequency for annealing temperature within the range (423-623) K, the deduced exponent s values, was found to decrease with increasing of annealing temperature through the frequency of the range (10^2-10^6) Hz. It was found that, the correlated barrier hopping (CBH) is the dominant conduction mechanism. Values of dielectric constant ε_1 and dielectric loss ε_2 were found to decrease with frequency and increase with temperature. The activation energies have been calculated for the annealed thin films.

Keywords: AC-conductivity chalcogenide thin films, dielectric constant, polarization.

INTRODUCTION

Chalcogenide materials have a great potential in phase-change random access memory (PRAM) devices are regarded to be a strongly promising candidate for the next generation nonvolatile memory (Doo Seok Jeong *et al*, 2011). These glasses are sensitive to the absorption of electromagnetic radiation and show a distinct photo induced effects. These properties show potential for numerous applications in active and passive optics (Mikhail Kibalchenko *et al*, 2010). Currently, amorphous materials can be electrically insulating, semiconducting or metallic in nature. Amorphous semiconductors have attracted immense interest in the field of phase change memory (PCMs) applications due to their electrical and optical properties and are potential candidates for many commercial applications. Phase change memories based on chalcogenide glasses have been found to be suitable candidates for replacing the conventional flash non-volatile random access memories (NVRAMS).They exhibit two types of switching behavior; threshold and memory switching, and this switching behavior is central to their technological importance and their uses as memories (Mohammad Mahbubur Rahman et al, 2013). The studying of ac-conductivity $\sigma_{a.c}$ (w) gives us information about the nature of the conduction mechanisms in a material. This conductivity $\sigma_{a.c}(\omega)$ can be represented by the empirical formula which follows a power low of the forms (A. M. Badr et al, 2011):

$$\sigma_{a.c}(\omega)=A\,\omega^2 \qquad ...(1)$$

Where, the frequency exponent lies in the range 0<s<1. The equation (1), can write the exponent s as follows

$$s=d(\ln \sigma(\omega)/d(\ln \omega)) \qquad ...(2)$$

Where ω: is the angular frequency.

A study of the frequency and temperature dependence of dielectric losses should provide information on relaxation processes. In the presence of relaxation effects, the dielectric constant may be given by:

$$\varepsilon^*(\omega)=\varepsilon_1(\omega)+i\varepsilon_2(\omega) \quad ....(3)$$

Where $\varepsilon_1(\omega)$: the real part of the complex dielectric (dielectric constant) and $\varepsilon_2(\omega)$ is the imaginary part(dielectric loss factor). The real part of dielectric constant could give by:

$$\varepsilon_1(\omega)=tC/a\ \varepsilon_o \quad ...(4)$$

Where (ε_o) is permittivity of the space (8.854x10^{-12}Fm-1),(a) is the effective electrode area, and (C) is the capacitance of the sample while the imaginary part is given by:

$$\varepsilon_2(\omega)= \sigma(\omega)/\varepsilon_o\ \omega \quad ...(5)$$

We can get $\sigma_{a.c}(\omega)$ by (M.I.Mohammed *et al*, 2002).

$$\sigma_{a.c^.}(\omega) =t/A.R \quad ...(6)$$

$$\tan\delta= \varepsilon_2(\omega)/ \varepsilon_1(\omega) \quad ...(7)$$

Where $\tan\delta$ is dissipation factor.

Describes the phase difference between the current and voltage with respect to the applied ac electric field. The dependence of a.c. conductivity $\sigma_{a.c^.}$ on temperature is usually obey the well-known relation (M.I.Mohammed et al, 2002):

$$\sigma_{a.c^.} = \sigma_{ï} \exp (- \Delta E / KT) \quad ...(8)$$

Where $\sigma_{ï}$, is constant, ΔE is the activation energy, T is the absolute temperature and K is Boltzmann constant.

Experimental parts

An alloy of $Ge_{10}Se_{20}Bi_{80}$has been prepared in evacuated quartz tube by the melt quenching technique from elements of high purity (99.999%). The $Ge_{10}Se_{20}Bi_{80}$ thin films deposited via the thermal evaporation method on glass substrate at room temperature of 3000& and rate of deposition (4.1) & /sec.The thin $Ge_{10}Se_{20}Bi_{80}$films were prepared in sandwich configuration between two A_l electrodes. For A.C. measurement we used HP.R2C unit, models 4274 A and 4275 A, and multifrequency LRC meter to generate a.c current in the frequency of the (10^2-10^6) Hz. The prepared thin films were annealed at 423,523 and 623K for 1/2 hour.

RESULTS and CONCLUSIONS

Present paper reports the frequency and temperature dependent of ac conductivity of $Ge_{10}Se_{20}Bi_{80}$ thin films in frequency of the range (10^2-10^6) Hz. Fig. (1) Show the variation of σ_{ac} conductivity (ù) as a function of frequency of $Ge_{10}Se_{20}Bi_{80}$thin films at various annealing temperature Ta=423,523 and 623K. This figure reveals that ln $\sigma_{a.c^.}(\omega)$ which represented in equation (1) increases linearly with ln(ù) at different annealing temperature in the range from 423K to 623K. The power law dependence of ac conductivity is of a universal nature and corresponds to short range hopping of charge carriers through trap sites separated by energy barriers of varied heights. Each pair of potential well corresponds to a certain time constant of transition from one site to another. We can notice from Fig.(1) that ac-conductivity dominates at higher frequency than at the lower values, which means that the $\sigma_{a.c^.}$ conductivities increases with increasing frequency. The phenomenon has been ascribed to relaxations caused by the motion of electrons or atoms. Such motion can involve hopping or tunneling between equilibrium sites (S.R. Elliott, 1987). The values of the frequency exponent s which presented in equation (2) were estimated from the slope of straight lines in Fig. (1).The annealing temperature dependence of the s for investigated films is shown in table (1).Table1 Represented the s values versus annealing temperature. It observed from table (1) that is a little difference between the calculated values of s, this may be due to the hopping process, also its values decreases with increasing annealing temperature. In most chalcogenide semiconductors the values of s from 0.7 to 1.0 at room temperature, and showed a tendency to decrease with increasing temperature. Therefore, the CBH has been extensively that the obtained experimental results agree with CBH in some values of annealing temperature of charge carriers across the defect state D+ and D- (H. E. Atyia) *et al*, 2008). According to Ghintini model (J.C.Giuntini *et al*, 1981), each pair of D+ and D- assumed to form a dipole with relaxation energy this type of energy can be attributed to the existence of a potential barrier over which the carriers hop (H. E. Atyia *et al*, 2008).

Frequency and temperature dependence of the dielectric constant ε_1 Dielectric analysis measures the electric properties of a material as a function of temperature and frequency. It measures two fundamental electrical characteristic of materials: the capacitive (insulating) nature, which represents its ability to store electric charge, and the conductive nature, which represents its ability to transfer electric charge. The variation of the dielectric constant ε_1 with frequency and temperature was studied for the investigated film. It is observed that ε_1 decreases with frequency and increases with temperature for the range of frequencies (100 Hz- 20000 Hz). After this range the behavior was changing to inverse behavior. The decrease in ε_1 with increasing frequency can be attributed to the contribution of the many components of polarization, electronic, ionic, dipolar or orientation and space charge. First, valance electrons relative nucleus. This type of polarization occurs at frequencies up to 10^{16} Hz. Second, ionic polarization occurs due to the displacement of negative and positive ions with respect to each other. The maximum frequency of ionic polarization is 10^{13} Hz. Third, dipole polarization occurs as a result of the presence of molecules with permanent electric dipole moments that can change orientation into direction of the applied electric filed. Dipole polarization occurs at frequencies up at about 10^{10} Hz. Finally, space-charge polarization occurs due to impedance mobile charge by interface. Space-charge polarization. Typically occurs at frequencies

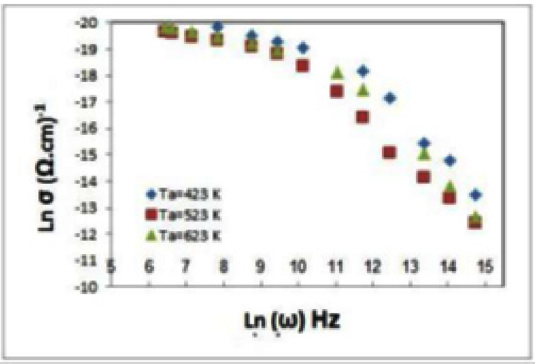

Fig. 1: Frequency dependence of σ_{ac} conductivity for annealed $Ge_{10}Se_{20}Bi_{80}$ thin films at various temperatures

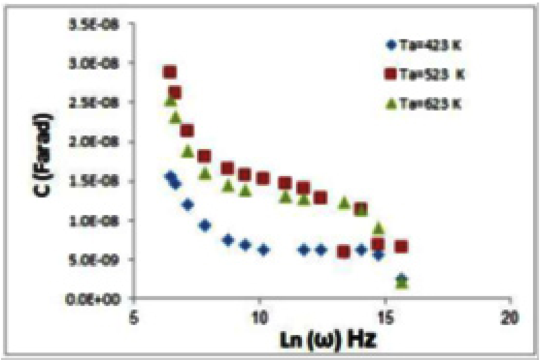

Fig. 2: Dependence of the capacitance on frequency at different annealing temperatures.

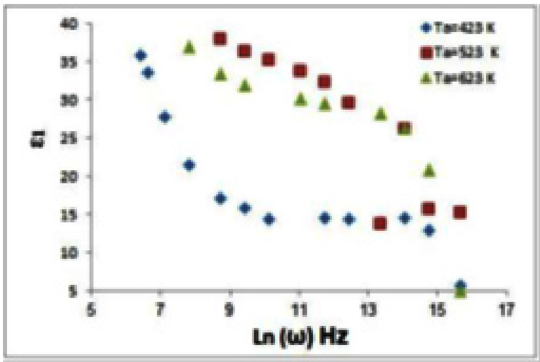

Fig.(3) represents the frequency dependence of the dielectric constant $\varepsilon 1$ for different annealing temperature

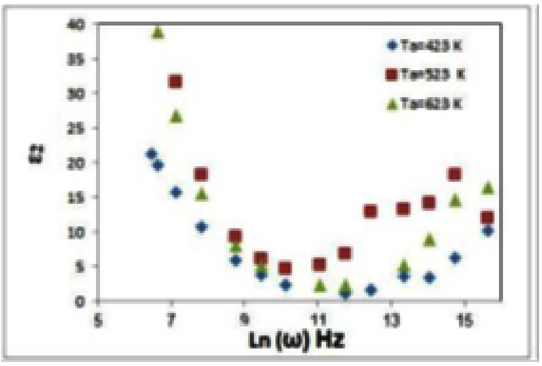

Fig. 4: Frequency dependence of the dielectric constant $\varepsilon 2$ (ω) for films at different annealing temperature

between 1 and 10^3 Hz. The total polarization of dielectric material can be represented by the sum of these four polarizations (M.W Barsoum, 1977).

Fig.(4) shows the frequency dependence of dielectric loss ε_2 at different annealing temperatures. it can be noticed from the figure that ε_2 decreases with increasing frequency and increases with temperature throughout the range (100Hz- 20000 Hz) of frequency and temperature, then the behavior change to inverse behavior. The increase of ε_2 with increasing temperature can be interpreted on the basis of the relaxation phenomena, which is divided into three parts; conduction loss, dipole loss and vibration loss. The losses that are attributed to conduction presumably involve the migration of ions over a large distance. This motion is the same as that occurring under direct current conditions. The ions jump over the highest barriers in the network. As the ions move, they impart some of the energy to the lattice as heat and the amount of heat lost per cycle is proportional to $\sigma_{a.c}(\omega)$ (J. M. Stevels, 1957) . At low temperature values, conduction losses, dipole losses and vibration losses have minimum values. However, at higher temperatures, conductions, dipole and vibration losses all contribute to the dielectric loss.

Table: 1

	Annealing temperature(K)		
	423 K	523K	623K
Capacitance(s)	0.922	0.901	0.873

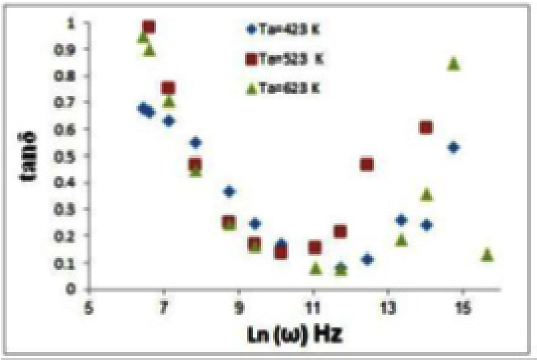

Fig. 5: Plot of tanδ against frequency at different annealing temperatures

This will lead to an increase in ε_2 with increasing temperature.

Fig.(5) shows the frequency dependence of loss tangent(tan δ) at different temperatures for a typical film $Ge_{10}Se_{20}$ Bi_{80} of thickness 230 nm. Tanδ has been found to increase with frequency at different temperatures, pass through a maximum value (tan δ) max and thereafter decreases. As the temperature is increased, the frequency at which (tanδ)max occurred shifted to higher frequencies. This type of behavior is similar to that described by Simmons (Safa's M. Hraibat et al, 2013) for films having parallel RC elements of the materials in series with a parallel combination of Schottky barriers capacitance and resistance.

Fig.6 Variation of ln $\sigma_{ac}(\omega)$ versus 1000/T for thin films $Ge_{10}Se_{20}$ Bi_{80} for different frequencies It is clear that at low temperatures and frequencies the conductivity is frequency dependent; however, with increasing temperature the conductivity becomes frequency independent, remaining almost constant throughout the entire frequency range. The strong temperature dependence may be due to the increase of the density of neutral defects Do (single-polaron states) at a higher temperature. The contribution toó ac (ω,T) from(CBH) of single polarons exceeds that of bipolarons at a high temperature (Sevi Murugavel and Manisha Upadhyay, 2011), i.e. the CBH of single polarons is then the predominant contribution to the AC conduction at higher temperature. The deduced values of activation energy reported in table (2) for various values of annealing temperature. The activation energy decreases with increasing frequency. The low value of the ac activation energy and the increase of $\sigma_{ac}(\omega)$, with the increase of frequency confirm that hopping conduction is the dominant current transport mechanisms. Thus, the increase of the applied frequency enhances

Table 2: Values of Ea.c(eV) as a function of frequency for fixed annealing temperatures

		Ea.c(eV)		
Temperature (K)		**423K**	**523K**	**623K**
Frequency	102Hz	0.1820	0.191	0.211
(Hz)	103Hz	0.1510	0.146	0.121
	106Hz	0.0420	0.02900	0..320

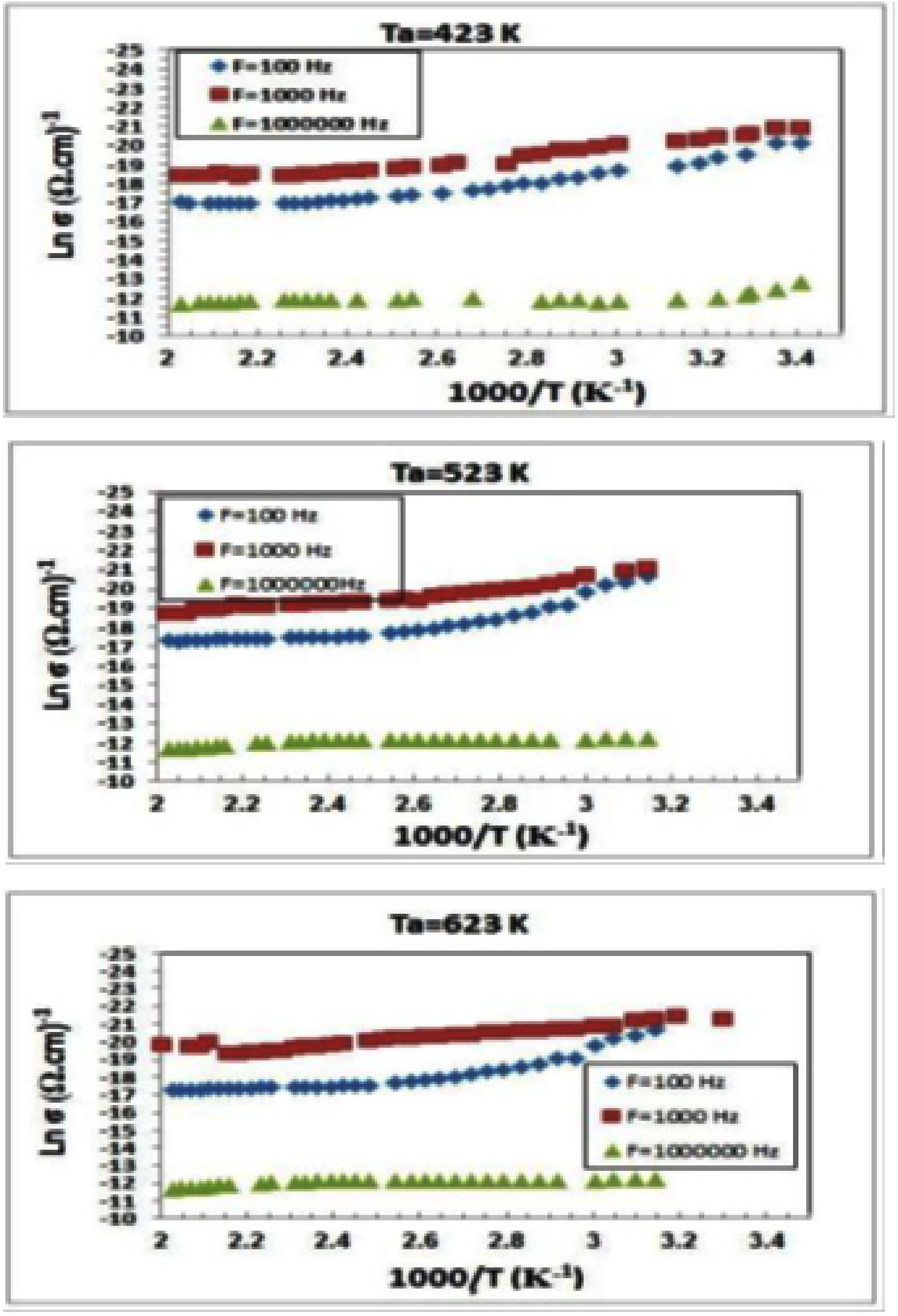

Fig. 6: Variation of ln σac (ω) versus 1000/T for thin films $Ge_{10}Se_{20}$ Bi_{80} for different frequencies

the electronic jumps between the localized states; consequently, the activation energy ΔE, decreases with increasing frequency (M.S. Hossaina et al, 2008).

CONCLUSIONS

Experimental results of the AC conductivity of the studied films seem to be frequency and temperature dependence. The values of the frequency exponent s are found to be temperature dependent, decreasing with increasing temperature. This confirmed the applicability of the CBH model. Both dielectric constant ε_1 and the dielectric loss ε_2 increased with temperature and decreased with frequency within the range (100 Hz-20000 Hz). The activation energy decreases with increasing frequency. And the increase of $\sigma_{ac}(\omega)$, with the increase of frequency confirm that hopping conduction is the dominant current transport mechanisms.

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